

## Theoretical Investigation on the Reactions of Dichlorosilane with Ammonia

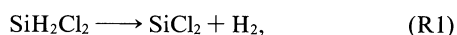
Shiro KOSEKI\*<sup>††</sup> and Akihiko ISHITANI<sup>†</sup>  
NEC Scientific Information System Development, Ltd.,  
34 Miyuki-ga-oka, Tsukuba 305

<sup>†</sup> VLSI Development Division, NEC Corporation,  
1120 Shimokuzawa, Sagamihara 229  
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$\text{SiH}_2\text{Cl}_2$  decomposition into  $\text{SiCl}_2$  and  $\text{H}_2$ ,  $\text{SiCl}_2$  insertions into  $\text{NH}_3$  and  $\text{SiH}_4$ , and 1,1-HCl elimination from  $\text{SiHCl}_2\text{NH}_2$ , which seem to occur in chemical vapor deposition processes of the silicon-nitride film growth, were theoretically investigated using second-order Moller–Plesset perturbation theory based on ab initio restricted Hartree–Fock method. The calculated results suggest that the Cl atoms create an energy barrier larger in Si–Si bond formation and facilitate the growth of stoichiometrically better films.

It is necessary to grow extremely thin silicon-nitride (SiN) films by chemical vapor deposition (CVD) processes in order to develop denser dynamic random access memory (DRAM) cells for the next generation of ultra-large scale integrated circuits (ULSIs).<sup>1)</sup> For this purpose, stoichiometrically better SiN films need to be produced with appropriate source gases under conventional CVD conditions. It has been proved experimentally<sup>2–4)</sup> that a mixture of  $\text{SiH}_2\text{Cl}_2$  and  $\text{NH}_3$  is more efficient than that of  $\text{SiH}_4$  and  $\text{NH}_3$  as a source gas for film growth. To our knowledge, the difference in the growth mechanism between these two cases has not been theoretically or experimentally researched in detail.

The film-growth processes from  $\text{SiH}_4$  and  $\text{NH}_3$  were examined using the experimental results and theoretical studies concerning reactions of  $\text{SiH}_4$  and  $\text{NH}_3$ .<sup>5)</sup> In the present study, we investigate the film-growth mechanism from  $\text{SiH}_2\text{Cl}_2$  and  $\text{NH}_3$ . Based on an analogy with the reactions of  $\text{SiH}_4$  and  $\text{NH}_3$ , the following four reactions seem to occur in a mixture of  $\text{SiH}_2\text{Cl}_2$  and  $\text{NH}_3$  at the CVD temperature:



Therefore, we carried out ab initio molecular orbital calculations on these four reactions; we discuss the effects of Cl atoms in comparison with the analogous reactions of  $\text{SiH}_4$  and  $\text{NH}_3$ . On the basis of the calculated results for these vapor-phase reaction, we also make some suggestions concerning the growth mechanism of SiN films when a mixture of  $\text{SiH}_2\text{Cl}_2$  and  $\text{NH}_3$  is adopted.

### Methods of Calculation

MC-311G(d,p)<sup>6)</sup> and 6-311G(d,p)<sup>7)</sup> basis sets were employed for the second-row atoms (Si and Cl) and for N and H atoms, respectively. The geometrical structures at stationary points were optimized within the restricted Hartree–Fock (RHF) approximation<sup>8)</sup> using the analytic gradient method.<sup>9)</sup> Vibrational analyses were also performed for all geometries. The relative energies of the geometries were recalculated using second-order Moller–Plesset perturbation theory (MP2)<sup>10)</sup> with an RHF zero-point energy correction. All of the calculations were made on an NEC SX2 machine using the quantum chemistry program code GAMESS developed by Schmidt et al.<sup>11)</sup>

### Results and Discussion

**Decomposition:** Reaction (R1) had been predicted to be the most favorable path of  $\text{SiH}_2\text{Cl}_2$  decomposition.<sup>12)</sup> The transition-state geometry **1** for (R1) is illustrated in Fig. 1; the energy barrier was calculated to be 80 kcal mol<sup>−1</sup> (see Fig. 2). Regarding an analogous decomposition of  $\text{SiH}_4$ , MP2 calculations yielded an energy barrier of 62 kcal mol<sup>−1</sup>; the energy barrier had been predicted to be 57 kcal mol<sup>−1</sup> in more reliable calculations.<sup>13)</sup> Therefore, since the MP2 calculations tend to slightly overestimate the energy barriers, the energy barrier of (R1) could be smaller than 80 kcal mol<sup>−1</sup>.

Gosavi et al.<sup>14)</sup> reported 36 kcal mol<sup>−1</sup> for the energy gap between the ground state (<sup>1</sup>A<sub>1</sub>) and the lowest triplet state (<sup>3</sup>B<sub>1</sub>) of  $\text{SiCl}_2$  using self-consistent-field methods; an energy gap in  $\text{SiH}_2$  had been predicted to be about 17 kcal mol<sup>−1</sup>.<sup>15)</sup> Since the large energy gap for  $\text{SiCl}_2$  is explained by the effect of electronegative Cl atoms,  $\text{SiCl}_2$  has been predicted to be less reactive than  $\text{SiH}_2$ .<sup>15)</sup> This prediction is consistent with the calculated results concerning the reversereaction of (R1) (a relatively large energy barrier of 46 kcal mol<sup>−1</sup>); a larger barrier may exist in  $\text{SiCl}_2$  insertions into  $\text{NH}_3$  and  $\text{SiH}_4$  (R2 and R3).

**Insertions:** The calculated results show that the energy barriers of (R2 and R3) are 21 and 25 kcal mol<sup>−1</sup>, respectively (Fig. 2). Raghavachari et al.<sup>16)</sup> and Gordon et al.<sup>17)</sup> reported an energy barrier of 13 kcal mol<sup>−1</sup> for

<sup>††</sup> Present address: Department of Chemistry, Faculty of Education, Mie University, 1515 Kamihama, Tsu 514.

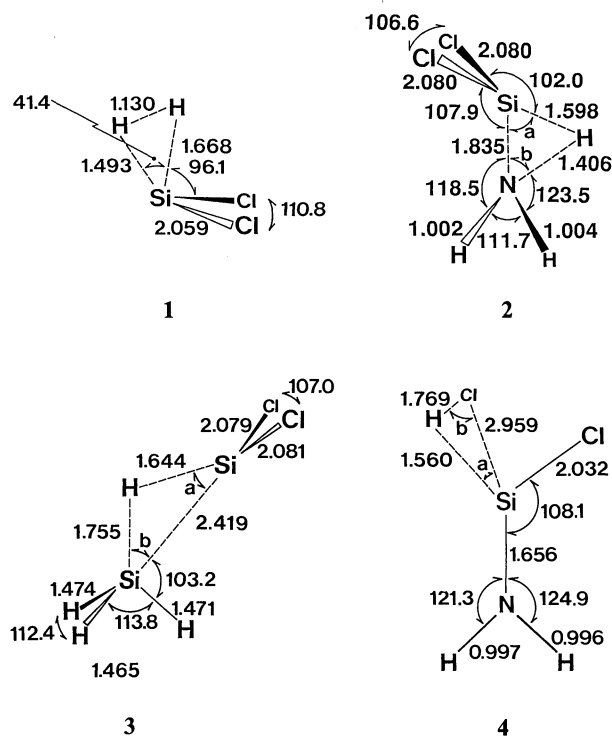


Fig. 1. Transition-state geometries of the thermal decomposition of  $\text{SiH}_2\text{Cl}_2$  **1**, the  $\text{SiCl}_2$  insertion **2** into  $\text{NH}_3$  ( $a=47.7$ ,  $b=52.3$ ), the  $\text{SiCl}_2$  insertion **3** into  $\text{SiH}_4$  ( $a=46.5$ ,  $b=42.8$ ), and the 1,1-HCl elimination **4** ( $a=29.2$ ,  $b=25.5$ , and  $\text{Cl-Si-Cl}=116.5$ ). The bond lengths and bond angles are in angstroms and degrees.

SiH<sub>2</sub> insertion into NH<sub>3</sub> and no energy barrier for SiH<sub>2</sub> insertion into SiH<sub>4</sub>. It was thus confirmed that SiCl<sub>2</sub> is less reactive than SiH<sub>2</sub>, as suggested above. The transition-state geometries (2 and 3 in Fig. 1) of (R2 and R3) have no symmetry since their C<sub>s</sub> geometries have 2 imaginary frequencies. An additional complex (5 in Fig. 3) having no negative force constant was found for (R2). The huge stabilization (22 kcal mol<sup>-1</sup>) of 5 would be caused by a strong interaction between the empty 3pπ orbital of SiCl<sub>2</sub> and the lone-pair orbital of NH<sub>3</sub>.<sup>18)</sup> The formation of Si–N bonds needs to be facilitated for the growth of SiN films, while Si–Si bonds cause the films to have some electric deficiencies.<sup>19)</sup> If the above discussion can be applied in a straight-forward way to reactions on the film surface, Si–Si bonds would be created much more easily than the Si–N bond when a mixture of SiH<sub>4</sub> and NH<sub>3</sub> is adopted. Then, the films tend to have a composition which deviated from the stoichiometrically correct one (Si<sub>3</sub>N<sub>4</sub>) and easily becomes Si-rich.<sup>3)</sup> On the other hand, the formation of Si–N bonds can be accelerated by a large NH<sub>3</sub> ratio in the mixture of SiH<sub>2</sub>Cl<sub>2</sub> and NH<sub>3</sub>,<sup>2,3)</sup> since both (R2 and R3) have similar energy barriers. Therefore, stoichiometrically better films are expected to be produced more efficiently using the latter mixture. This theoretical interpretation thus seems to reasonably explain the

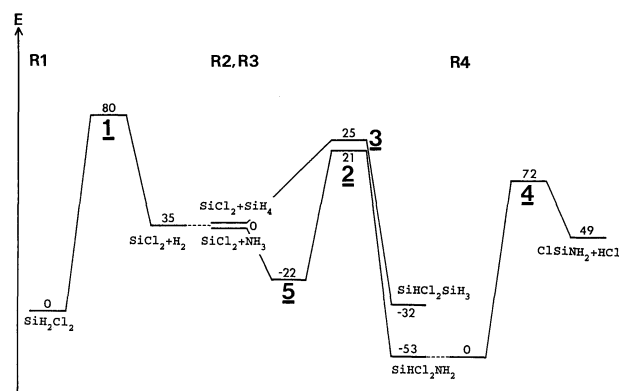


Fig. 2. Reaction profile for R1, R2, R3, and R4. The reactions proceed from left to right. Bold numbers correspond to those in Figs. 1 and 3. The relative energies are in kcal mol<sup>-1</sup>.

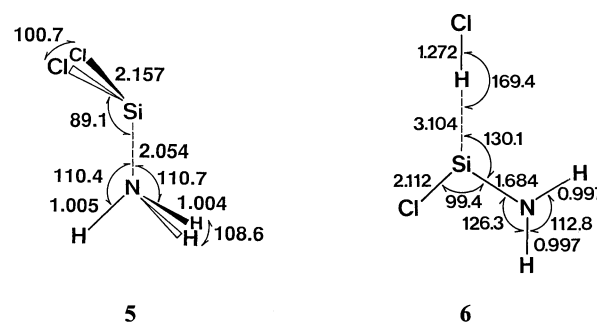


Fig. 3. Geometries of additional complexes for the  $\text{SiCl}_2$  insertion **5** into  $\text{NH}_3$  and the 1,1-HCl elimination **6**. The bond lengths and bond angles are in angstroms and degrees.

experimental results.<sup>2,3)</sup>

**Elimination:** The reaction energy ( $49 \text{ kcal mol}^{-1}$ ) of the 1,1-HCl elimination (R4) from  $\text{SiHCl}_2\text{NH}_2$  is much smaller than that ( $129 \text{ kcal mol}^{-1}$ ) of 1,1- $\text{Cl}_2$  elimination and some kinds of 1,2-eliminations ( $75\text{--}78 \text{ kcal mol}^{-1}$ ). The energy barrier for R4 was calculated to be  $72 \text{ kcal mol}^{-1}$  (Fig. 2); its transition-state geometry **4** is depicted in Fig. 1. This barrier is somewhat larger than that ( $67 \text{ kcal mol}^{-1}$ )<sup>20)</sup> of the 1,1- $\text{H}_2$  elimination from  $\text{SiH}_3\text{NH}_2$ . Then, the product  $\text{Cl-Si-NH}_2$  may react with another  $\text{NH}_3$  to create a new Si-N bond. Accordingly, the repeat of  $\text{SiCl}_2$  insertion into an N-H bond and HCl elimination could produce various SiN particles.

An additional complex (**6** in Fig. 3), whose force constant matrix is positive definite, is predicted to be formed before (R4) is completed. Although highly polarizes HCl seems to strongly interact with the lone-pair of the Si atom in Cl-Si-NH<sub>2</sub>, **6** is only about 1 kcal mol<sup>-1</sup> lower than the dissociation limit, in contrast with that of **5**. This small stabilization energy might be explained in the following manner: More electronegative

groups (Cl and NH<sub>2</sub>) in Cl-Si-NH<sub>2</sub> derive a large positive charge (+0.7 e) on the Si atom, so that the attractive interaction between the H atom and the lone-pair of the Si atom is weakened by the charge-charge repulsion between the H atom and the entire Si atom. This fact would also explain an energy barrier of 23 kcal mol<sup>-1</sup> for the reverse reaction of (R4).

### Conclusion

The present work involved four vapor-phase reactions (R1—R4) using ab initio molecular orbital methods. It is preliminarily concluded that Cl atoms create an energy barrier that is larger in Si-Si bond formation, which seems to facilitate the growth of stoichiometrically better films. We plan to investigate fundamental reactions on the film surface based on the calculated results for these vapor-phase reactions.<sup>21)</sup>

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